

ris Publishers

Research Article

Copyright © All rights are reserved by Yohichi Kohzuki

Dislocation-Monovalent Cations Interaction in Sodium Halide Crystals During Plastic Deformation

Yohichi Kohzuki*

Department of Mechanical Engineering, Saitama Institute of Technology, Japan

*Corresponding author: Yohichi Kohzuki, Department of Mechanical Engineering, Saitama Institute of Technology, Japan Received Date: December 13, 2023 Published Date: January 08, 2024

Abstract

The combination methods of strain-rate cycling tests and ultrasonic oscillation were conducted during the plastic deformation of NaCl:Li⁺ (0.5 mol% in melt) single crystals at 77 to 253 K. Assuming that the temperature dependence of τ_p is approximated to the Cottrell-Bilby relation taking account of the Friedel relation, the activation energy (G_0) for overcoming the isotropic defect ($\Delta \epsilon$) around Li⁺ ion by a dislocation is somewhat large as compared with NaBr:Li⁺ crystal. τ_p is the effective stress due to the dopants, which was obtained by the above-mentioned method. This difference in G_0 value between the two kinds of crystals is caused by the result that $\Delta \epsilon$ becomes slightly large in the sodium halide crystal when anion size of host crystal is approached to that of the Na⁺ ion (i.e., $\Delta \epsilon$ varies from 0.079 for NaBr:Li⁺ to 0.089 for NaCl:Li⁺).

Keywords: Mobile dislocation; Isotropic defect; Effective stress; Cottrel-Bilby model; Ultrasonic oscillation

Introduction

Combination method of strain-rate cycling tests and ultrasonic oscillation has been considered to give the information on the dislocation motion on the slip plane containing many weak obstacles (e.g., impurity atoms, isolated and clustered point defects) and a few strong ones (forest dislocations) in bulk during plastic deformation at low temperature (e.g., [1]). When the ultrasonic oscillatory stress is applied to a plastic deforming crystal, the applied stress drop is appeared on a stress-strain curve. The decrease of applied stress increases with stress amplitude at a given temperature and shear strain [1]. The strain-rate sensitivity of flow stress, which is inversely proportional to the average length of dislocation segments [2], was measured here on the basis of stress change due to the strain-rate cycling in the middle of ultrasonic oscillatory stress. By using the above-mentioned method, the influence of anion size of host crystal on the mechanical properties is investigated at low temperature for sodium halide crystals doped with monovalent cations (NaCl:Li⁺ and NaBr:Li⁺) in this paper.

Experimental procedure

NaCl single crystals doped with Li⁺ ions were prepared by cleaving to the size of 5×5×15 mm³ as the specimens used in this study. The concentration of dopant ions was 0.5 mol% in melt. The specimens were annealed at 973 K for 20 h and were gradually cooled to room temperature at 40 Kh⁻¹ for the purpose of reducing dislocation density as much as possible.

The strain-rate cycling tests combined with ultrasonic oscillation (20 kHz) were conducted keeping the stress amplitude (τ_v) constant during plastic deformation for the specimens at 77 to 253 K. Application or removal of ultrasonic oscillation causes the stress drop ($\Delta \tau$) and strain-rate cycling between $\dot{\epsilon}_1(1.1 \times 10^{-5} \text{ s}^{-1})$ and $\dot{\epsilon}_2$ (5.5×10⁻⁵ s⁻¹) makes the stress change ($\Delta \tau$ ') during the tests. The strain-rate sensitivity (λ) of flow stress was derived from $\Delta \tau$ ' value (i.e., $\lambda = \Delta \tau' / \Delta \ln \dot{\epsilon} = \Delta \tau' / 1.609$). This was detailed in the previous paper [3].

Results and Discussion

The variation of $\Delta\tau$ with shear strain is shown in Figure 1 for NaCl:Li* (0.5 mol%) at 143 K. The numbers besides each symbol in the figure represent the output voltage from the piezoelectric transducer set on the bottom of specimen, which is proportional to $\tau_{\rm v}$. $\Delta\tau$ values do not largely change at all stress amplitudes and are

almost constant, irrespective of strain. $\Delta \tau$ also tends to be larger with τ_v at a given temperature and strain. While the variation of λ of flow stress with it is shown in Figure 2 for the same specimen. λ values tend to increase with strain above about 5% and decrease with τ_v at a given temperature and strain. $\Delta \lambda / \Delta \tau_v$ is small at a given strain for low and high τ_v regions, as seen in Figure 2.



Figure 1: Variation of the stress decrement (ΔT) due to the superimposition of ultrasonic oscillation with shear strain (ϵ) at 143 K and various stress amplitude (r_v) for NaCl:Li⁺ (0.5 mol%). The numbers besides each symbol represent the output voltage from the piezoelectric transducer set on the bottom side of specimen.





Modern Concepts in Material Science

The relations between $\Delta \tau$ and λ for each output voltage in Figures 1 and 2 are plotted for the specimens at strains of 5 and 8% and 143 K in Figure 3. The $\Delta \tau$ versus λ curves have stair-like shape: two bending points and two plateau places. That is to say, the first

plateau place ranges until first bending point and second one extends from the second bending point. λ decreases with $\Delta \tau$ between the two bending points.





Figure 4 shows the $\Delta \tau$ versus λ curves for NaCl:Li^{*} (0.5 mol%) at various temperatures. The numbers besides each symbol in the figure represent the temperature of the specimens. The curves also have stair-like shape at low temperatures (i.e., 77 and 193 K), but the first plateau place cannot be seen at 253 K, as can be seen in Figure 4. Similar variation of $\Delta \tau$ versus λ curve with temperature has been observed for NaBr:Li^{*} (0.5 mol%) single crystals at 77 to 263 K [4]. Figure 5 concerns the relative curves for NaBr:Li^{*}. The length of $\Delta \tau$ within first plateau place is referred as τ_p in Figure 5. It seems from Figures 4 and 5 that τ_p tends to be lower at higher

temperature for the two kinds of specimens: NaCl and NaBr crystals doped with Li⁺ ions. When a dislocation moves overcoming the short-range obstacles such as dopant ions with thermal activation during plastic deformation, it is considered that τ_p is the effective stress due to the dopants, because τ_p depends on the type and the density of obstacles (e.g., [5, 6]). Therefore, observation of τ_p will provide information on dislocation-monovalent cations interaction in the specimens. Namely, τ_p versus temperature relation reflects the dislocation-monovalent cations interaction in a crystal.





Isotropic defects ($\Delta\epsilon$) occur around Li* ions in NaCl:Li* or NaBr:-Li* crystals, since the dopant ionic size is different from the substitutional ions (Na*) of the host crystal. Then, the force-distance profile between a dislocation and the symmetry defect is expressed by the Cottrell and Bilby relation [7] taking account of the Friedel relation [8]. The plots in Figures 6 and 7 represent τ_p values for NaCl:Li* and NaBr:Li* crystals at 77 to 263 K and are approximated by the Cottrell and Bilby relation, where the solid curves in these figures were obtained by numerical calculation with the parameters ($\tau_{p0'}$, T_{r'} and G_0) as to be fitted to the solid circles using the least-squares

method. These parameters are denoted in Table 1. $\tau_{\rm p0}$ is the $\tau_{\rm p}$ value at the temperature of 0 K. $T_{\rm c}$ is the critical temperature at which $\tau_{\rm p}$ is zero and does not depend on the concentration of dopant ions [9]. $\tau_{\rm p}$ value decreases with increasing temperature and approaches to zero at the critical temperature (T_c) as seen in Figures 6 and 7. $T_{\rm c}$ for NaCl:Li^+ crystal is near the melting point and is much higher than $T_{\rm c}$ (469 K) for NaBr:Li* [10, 11]. G_0 is the activation energy for the breakaway of a dislocation from the isotropic defect around dopant (i.e., Li* ion here) in the absence of an applied stress.

Table 1: Parameters used for calculation.

Specimen	$ au_{p0}(Mp_a)$	<i>G</i> ₀ (eV)
NaCl:Li* (0.5 mol% in the melt)	0.57 [11]	0.55 [11]
NaBr:Li⁺ (0.5 mol% in the melt)	1.08 [10]	0.50 [10]

Table 2: Values of the lattice constants at room temperature (298 K) and atmospheric pressure, and defect size (Δε) for each specimen.

Crystal	Lattice Constants (Å)	Specimen	Δε
NaCl	5.64 [12]	N=CLI :+	0.089
LiCl	5.14 [12]	Nacit	
NaBr	5.97 [12]	No Dec Lit	0.070
LiBr	5.50 [12]	Nabr:Ll'	0.079



Figure 6: Dependence of τ_{p} on temperature for NaCl:Li⁺ (0.5 mol%).



The values of $\Delta\epsilon$ are given in Table 2 for each specimen. These values are calculated from the difference between the lattice constants of host crystal and dopant one. The lattice constants are also given in the table. As a result, the $\Delta\epsilon$ is 0.089 for NaCl:Li⁺ crystal and is 0.079 for NaBr:Li⁺. As the anion size of host crystal is slightly small, the isotropic defect around Li⁺ ion is somewhat large. This leads to the difference in activation energy between the two kinds of crystals (G₀=0.55eV for NaCl:Li⁺ [11], G₀=0.50eV for NaBr:Li⁺ [10]) for overcoming the isotropic defect by a dislocation.

Conclusion

 $\Delta \tau$ versus λ curves, which were obtained by the strain-rate cycling tests combined with ultrasonic oscillation, have stair-like shape for NaCl:Li⁺ (0.5 mol%) single crystal at low temperature. Observation of $\Delta \tau$ at the first bending point on the curve, i.e. τ_{n} ,

gives us useful information on moving dislocation during plastic deformation. Assuming that the temperature dependence of $\tau_{\rm p}$ is approximated to the Cottrell-Bilby relation taking account of the Friedel relation, the activation energy is 0.55 eV for overcoming the isotropic defect around the dopant (Li⁺) in NaCl:Li⁺ crystal by a dislocation. As the anion size of host crystal is small, its value becomes somewhat large in contrast to 0.50 eV for NaBr:Li⁺, since the isotropic defect around Li⁺ ion becomes slightly large in the crystal (i.e., $\Delta\epsilon$ varies from 0.079 for NaBr:Li⁺ to 0.089 for NaCl:Li⁺).

Acknowledgement

Dr. T. Ohgaku, as well as S. Yamaguchi are acknowledged for his collaboration in the analysis on $\Delta \tau$ and λ data, as well as for his experimental assistance.

Conflict of Interest

No conflict of interest.

References

- 1. Kohzuki Y (2018) Study on dislocation-dopant ions interaction in ionic crystals by the strain-rate cycling test during the Blaha effect. Crystals 8(1): 31-54.
- 2. Conrad H (1964) Thermally activated deformation of metals. Journal of Metals 16: 582-588.
- Kohzuki Y (2023) Study on X-ray-Induced defects as barriers for dislocation motion by strain-rate cycling tests combined with ultrasonic oscillation. Modern Concepts in Material Science 5(2).
- Kohzuki Y, Ohgaku T (2017) Study on influence of additive and host ions on deformation characteristics of alkali halide crystals by strain-rate cycling tests during the Blaha effect. Crystal Research and Technology 52(3): 1600290.
- Ohgaku T, Matsunaga T (2009) Interaction between dislocation and divalent impurity in KBr single crystals. IOP Conference Series: Materials Science and Engineering 3(1): 012021.

- Ohgaku T, Teraji H (2001) Investigation of interaction between a dislocation and a Br⁻ ion in NaCl:Br⁻ single crystals. Phys. Status Solidi A 187(2): 407-413.
- Cottrell AH, Bilby BA (1949) Dislocation theory of yielding and strain ageing of iron. Proceedings of the Physical Society of London A 62: 49-62.
- 8. Friedel J (1964) Dislocations. Pergamon Press, Oxford pp. 223-226.
- Ohgaku T, Takeuchi N (1992) Interaction between a dislocation and monovalent impurities in KCl single crystals. Phys. Status Solidi A 134(2): 397-404.
- 10. Kohzuki Y (2022) Study on thermal activation barrier for dislocation motion by strain-rate cycling tests combined with ultrasonic oscillations. Modern Concepts in Material Science 5(1).
- 11. Kohzuki Y, Ohgaku T (2016) Activation energy for dislocation breakaway from various monovalent ions in NaCl single crystals. Philosophical Magazine 96(29): 3109-3119.
- 12. Sirdeshmukh DB, Sirdeshmukh L, Subhadra KG (2001) Alkali halides. Hull R, Osgood RM, Sakaki Jr.H, Zunger A, eds., Springer-Verlag, Berlin Heidelberg p. 6.