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Review Article

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Experimental Verification of the Mechanical State Equation Adequacy for Description of the Creep of Some Polymer Materials under Non-stationary Modes of Static and Cyclic Loading in Plain Stress

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Abstract

The equation of mechanical condition for the description of deformation of polymer and polymer composite materials under complex modes of long static and cyclic loading in condition of the plain stress state is developed. The experimental data show its efficiency. The means of accounting the influence of natural aging and the damage accumulation data of a material are suggested. The opportunity of using of a simplified variants of the equation is shown.

Keywords: Creep; Polymer materials; Mechanical state equation; Complex modes of loading; Experimental test

Introduction

Polymer and composite polymer materials proved an important class of structural materials with rheonomous properties. A variety of approaches to evaluation of the deformation of these materials and the time to failure can be found in the literature [1]. In general case the total strain under continuous loading is defined by the sum of four components: instantaneous-elastic, instantaneous -plastic, visco-elastic, and visco-plastic [2]:

$$e_j = e'_j + e''_j + e'''_j + e'''_j$$
 (1)

The components of the total strain deviator (Eq.1) are represented as:2 $% \left({{\left[{{{\rm{c}}_{\rm{m}}} \right]}_{\rm{m}}} \right)$

$$e_{j} = \frac{1+\nu}{E(\sigma_{i})} + \frac{3}{2} \int_{\lambda} \frac{d_{i}^{*}}{\sigma_{i}} S_{j} + \int_{0}^{\tau} \frac{d}{d\theta} \sigma_{j} R(\tau-\theta) d\theta + \int_{0}^{\tau} \frac{d}{d\theta} \sigma_{j} \Omega(\tau-\theta) (2)$$

or as:

$$e_{j} = \frac{1+\nu}{E(\sigma_{i})}S_{j} + \frac{2}{3}\int_{\lambda}\frac{S_{j}}{\sigma_{i}}\frac{d\sigma_{i}}{E_{k}} + \int_{0}^{\tau}d\xi\int_{0}^{\xi}\frac{d\sigma_{j}}{d\theta}K(\tau-\theta)d\theta + \int_{0}^{\tau}d\xi\int_{0}^{\xi}\frac{d\sigma_{j}(\theta)}{d\theta}B(\tau-\theta)d\theta$$
(3)

where S_j is the deviatorial stress components; $\boldsymbol{\nu}$ is the Poisson's ratio; \mathbf{E} – modulus of elasticity (in general case the non-linear elasticity should be considered); is the von-Mises equivalent stress, $\sigma_i = \sqrt{3/2}\sqrt{J_2^S}$; J_2^S is the second invariant of deviatorial stress; $\tilde{\sigma}_{ij} = f(\sigma_{ij})$ are the components of a characteristic stress; $K(\tau - \theta)$ and $R(\tau - \theta)$ are the cores of after action, $\tau - \theta = \xi$ is the time (it can be actual or dimensionless time); $\Omega(\tau - \theta)$ and $B(\tau - \theta)$ are the time functions; de_i^* is the equivalent strain increment, $e_i = \sqrt{2/3}\sqrt{\sum_{y} e_y^2}$, $de_i^* = dL\sqrt{2/3}$; E_k is the tangential modulus, $E_k = d\sigma_i/de_i$, $L = \int \sqrt{\sum_{y} (de_y)^2}$ is the plastic strain path, Odlvist's parameter.

For calculation of the relative change of a unit volume we suggest using a characteristic strain:

$$\varepsilon_{\nu} = 3(1-2\nu)\sigma_0/E(\sigma_i)$$
 , (4)

where $\sigma_0 = (\sigma_1 + \sigma_2 + \sigma_3)/3$ is the hydrostatic stress.

The non-linear law of instantaneous elasticity was defined in testing materials at fast (instant) unloading of specimen from the given level of stress attained at a high enough rate. To find out the dependence of instantaneous plastic component of the total strain on the stress the same tests were used, but the residual strains were measured after unloading and after holding a certain time in unloading. The instantaneous plastic strain component can be taken into account within the range of the visco-plastic component.

In description of the rheonomous components of the total strain the processes of shear are considered on condition of the volume stability of material. The visco-elastic component of strain is determined as:

$$e_j^{"} = \frac{3}{2} \int_0^{\tau} f_1(\frac{\sigma_i}{\sigma^*}) \varphi_1[\mu_s(\theta)] \frac{S_j}{\sigma_i} K(\tau - \theta) d\theta , (5)$$

where μ_s is the Lode's factor, σ^* is the dimensional constant. The function φ has to be obtained by testing specimens at the complex stress state.

To define the visco-plastic strain component e_j^{m} is necessary to deduce (during the whole period of stationary mode of loading) the appropriate instantaneous and visco-elastic components from the total strain it. The visco-plastic component is described as follows from [3 - 4]:

$$e_{j}^{""} = \frac{3}{2} \int_{0}^{\tau} f_{2}(\frac{\sigma_{i}}{\sigma}) \varphi_{2}[\mu(\theta)] \frac{S_{j}}{\sigma_{i}} B(\tau) d\tau$$
(6)

The calculation of strain components according to (2) and (3) at any unsteady loading condition and known constants is reduced to quadratures. The evaluation of the loading regime relative to the specified path of deformation (the stress relaxation under constant deformations should be considered as well) is carried by applying the step-by-step procedure in a general case. The constant and functional parameters of equations of mechanical state of polymer composite materials essentially depend on temperature. Methods of testing equipment and characteristics of the samples are presented in [1-13]. If only temporal parameters of the equation depend on temperature, then under some additional conditions the temperature-temporary analogy may be established, on which the process of no isothermal loading is reduced to the isothermal one

within the given time. Within the range of physical nonlinearity of material on the temperature depend not only temporal, but also the force parameters of equations (2) and (3).

In such cases a method of transformation of a stepwise non isothermal mode of loading to an equivalent isothermal mode is developed. The finite magnitudes of visco-elastic and visco-plastic components of the total strain are obtained by superposition of a series of loading impulses applied at actual or reduced moments of time.

The natural aging (aging time τ_c) and damage (Π) under continuous loading affect the process of creep of material and the ultimate strain [14]. The accumulated strains and damages change the relationship between the components of the total strain at step-wise loading regimes (with partial or complete intermediate unloading). At long pauses after loading a number of concurrent processes develops in material, in particular, damage remedies and natural aging. Equations (2), (3) contain functional parameters (depending on τ_c and Π) [15], which should be obtained by analysing the additional tests results. The account for the damage measure Π appears effective only by the description of strains directly preceding the moment of failure.

The obvious may be the application of the suggested equation of the mechanical states to the conditions of the cyclic stress state [16]. The instantaneous components of the total strain are calculated considering the maximal ones through the stress cycle. The viscoelastic component can be calculated directly by substituting the variable stress. A number of approaches is suggested for calculation of the visco-plastic strain component [4]. The possibility of application of the simplified variant of the equation of mechanical states was considered. The process of deformation (creep) of tubular specimens of aged polyethylene (14 years) under various regimes of stationary and non-stationary loading by axial tensile load, internal pressure and torque may be described with sufficient accuracy by the accounting for the instantaneous elastic and rheonomous components of the total strain:

$$e_{j} = \frac{1+\nu}{E(\sigma_{i})}S_{j} + \frac{3}{2}\int_{0}^{\tau} f(\frac{\sigma_{i}}{\sigma^{*}})\varphi[\mu_{s}(\theta)] \frac{S_{j}}{\sigma_{i}}N(\tau-\theta)d\theta$$
⁽⁷⁾

The calculated strains deviate from the experimentally obtained ones by 20%, and under the loading conditions without torque the difference does not exceed 13%. The examples of experimental and calculated creep diagrams are given in Figures 1-6. The figures show the qualitative and quantitative differences between experimental creep curves and analysis (description) of figures is not necessary (Figures 1-6).











Figure 3: Non-stationary static loading sample and creep diagrams of HDPE (n = 0.5) 1 – experimental curves; 2 – calculated curves.

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Summary

The above conclusions also can be extended on the cases of cyclic alteration of all or any one stress component. However, it is indicative that despite of nonlinearity of the genetic stress equation, the evaluation of strain components can be carried out considering the average stress magnitudes over the loading cycle. Analysis [17] shows that the rigorous account of the cyclic stress variation results in insignificant refinement of strain components [18–21]. In this paper above equations are applicable to a class of semi-crystalline polymeric materials such as low-density polyethylene, PVC and PTFE 4 and 4D [22–25].

Acknowledgment

None.

Conflict of Interest

No conflict of interest.

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